NATIONAL ADVISORY COMMITTEE FOR AERONAUTICS

TECHNICAL NOTE 2862

INFLUENCE OF NONMARTENSITIC TRANSFORMATION PRODUCTS ON MECHANICAL PROPERTIES OF TEMPERED MARTENSITE

By J. M. Hodge and W. T. Lankford

United States Steel Company

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INFLUENCE OF NONMARTENSITIC TRANSFORMATION PRODUCTS ON

MECHANICAL PROPERTIES OF TEMPERED MARTENSITE

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SUMMARY

The influence of nonmartensitic transformation products on the mechanical properties of tempered martensite is presented for samples of an SAE 4340 steel, partially isothermally transformed to specific high-temperature transformation products and quenched and tempered to hardness values of from 25 to 40 Rockwell C. The effects of upper bainite in amounts of 1, 5, 10, 20, and 50 percent, of 5 percent ferrite, and of 5 percent pearlite on the tensile, impact, and fatigue properties are evaluated.

INTRODUCTION

This study was undertaken to evaluate the influence of the presence of definite percentages of specific high-temperature transformation products on the mechanical properties of tempered martensite. Such information is obviously desirable in order to apply intelligently the results of hardenability measurements to the selection of steels for specific purposes. Although the results obtained are in terms of transformation products formed isothermally, it is believed that they can be applied to the mixed microstructures ordinarily obtained on continuous cooling, since the general nature of the nonmartensitic products obtained in such cases will usually be known.

This investigation was sponsored by the Chrysler Steel Standardization Committee which consists of the following men: M. F. Garwood, Chairman, Chrysler Corp.; M. W. Dalrymple, Bethlehem Steel Corp.; M. J. Day, United States Steel Corp.; M. Grossmann, United States Steel Corp.; R. B. Hooper, Carnegie-Illinois Steel Corp.; E. Iarned, Youngstown Sheet & Tube Co.; D. H. Ruhnke, Republic Steel Corp.; E. T. Walton, Crucible Steel Co. of America; R. L. Wilson, The Timken Roller Bearing Co.; and J. R. Zanetti, Great Lakes Steel Corp. The report has been made available to the National Advisory Committee for Aeronautics for publication because of its general interest.

MATERIALS AND EXPERIMENTAL WORK

An SAE 4340 steel of the following composition was used in this investigation:

Material in the form of 7/8-inch-diameter rolled bars was furnished to the various participating laboratories by the Carnegie-Illinois Steel Corp. These bars were reforged to 1/2-inch squares for impact testing and to 5/8- to 3/4-inch rounds for tension and fatigue tests. Samples for impact testing were heat-treated in the form of 0.420-inch-square bars and then machined to standard V-notch Charpy specimens. Tension and fatigue specimens having a diameter of 0.25 inch were used; these were machined 0.020 inch oversize for heat treatment and finished to final size after heat treatment. All material was normalized from 1650° F and tempered for 1 hour at 1200° F before machining.

The tempered martensite samples were austenitized at 1600° F for 1 hour, oil-quenched, and tempered at 800°, 1000°, and 1200° F for impact testing and at 800°, 900°, 1000°, 1100°, and 1200° F for tension and fatigue testing. The samples containing nonmartensitic products were austenitized at 1600° F for 1 hour, transferred to a salt or metal bath, and held in accordance with the schedule outlined below. They were then oil-quenched and tempered 1 hour at 800°, 900°, 1000°, 1100°, and 1200° F. The hardness values for the samples ranged from 25 to 40 Rockwell C.

ISOTHERMAL TREATMENTS

Micro	structure	Temperature	Time
5 percent 10 percent 20 percent		850° F 850° F 850° F 850° F 1175° F	15 seconds 55 seconds 90 seconds 115 seconds 25 minutes 10 minutes 20 minutes
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The 5-percent-pearlite structure contained approximately 8 percent free ferrite. Wherever the term "5 percent pearlite" is used, it will be understood that reference is made to a structure containing 5 percent pearlite and 8 percent ferrite.

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The impact studies on all structures were carried out at the Duquesne Works of the Carnegie-Illinois Steel Corp. The as-quenched microstructures of the tension and fatigue test specimens, as heat-treated at each of the cooperating laboratories, were checked and rated metallographically by Mr. M. A. Orehoski at Duquesne. The tension and fatigue tests were carried out in the cooperating laboratories as follows:

Microstructure

Laboratory

Tempered martensite	Crucible Steel Co. of America
l percent bainite	Chrysler Corp.
5 percent bainite	Republic Steel Corp.
10 percent bainite	Bethlehem Steel Corp.
20 percent bainite	American Steel & Wire Co.
50 percent bainite	The Timken Roller Bearing Co.
5 percent ferrite	Great Lakes Steel Corp.
5 percent pearlite	Carnegie-Illinois Steel Corp., South Works

RESULTS AND DISCUSSION

The experimental results are tabulated and presented graphically as follows: Impact properties, table I and figures 1 and 2; tensile properties, table II and figure 3; fatigue properties, table III and figure 4; and ratings of mechanical properties, table IV.

The effects of upper bainite are summarized graphically in figure 5 in the form of a plot of the mechanical properties as a function of the percentage of upper bainite at a constant hardness of 34 Rockwell C. It will be noted from this summary that upper bainite exerts its major effects on the impact properties and yield strength or yield-tensile ratio. The effect on impact properties was, as might be expected, more pronounced in tests conducted at -40° F than at room temperature. The impact strengths are, in general, only slightly decreased by upper bainite in amounts up to 5 percent or by 5 percent ferrite, but are significantly decreased by 10 to 20 percent upper bainite and very markedly lowered by 50 percent upper bainite or 5 percent pearlite.

The tensile properties are, in general, much less sensitive to the presence of nonmartensitic products than are the impact properties. The yield-tensile ratio, however, is quite markedly lowered by the presence of 50 percent upper bainite and seems to be somewhat affected by upper bainite in amounts ranging from 5 to 20 percent or by 5 percent of pearlite. These results also indicate somewhat lower values in reduction of area for material containing 50 percent upper bainite, but no significant effect of smaller percentages. The tensile strength and elongation values

for a given hardness seem not to be significantly affected by microstructural changes over the range studied.

The fatigue properties also seem generally insensitive to the presence of upper bainite in amounts below 50 percent, but are significantly lowered by the presence of 50 percent upper bainite or 5 percent pearlite.

The relatively large effect of 5 percent pearlite on toughness, as reflected by the impact strengths at a given hardness, is noteworthy, as the nonmartensitic products in plain carbon steels and some low-alloy steels are predominantly pearlitic, and such steels might, therefore, be expected to have poorer toughness when slack-quenched than alloy steels in which the nonmartensitic products would be predominantly bainitic.

SUMMARY OF RESULTS

The influence of nonmartensitic transformation products on the mechanical properties of tempered martensite was evaluated on samples of an SAE 4340 steel, partially isothermally transformed to specific high-temperature transformation products and quenched and tempered to hardness values of from 25 to 40 Rockwell C. The results may be summarized as follows:

- 1. For a given hardness value, the impact properties were significantly lowered by the presence of upper bainite in amounts of 10 percent or above or by 5 percent of pearlite, but were not significantly affected by 1 to 5 percent of upper bainite or by 5 percent of ferrite.
- 2. The tensile ductility remained relatively unaffected by the high-temperature transformation products, but the yield-tensile ratio was markedly lowered by the presence of 50 percent upper bainite or 5 percent pearlite.
- 3. The fatigue properties were likewise lowered somewhat by the presence of 50 percent upper bainite or 5 percent pearlite, but were not significantly affected by the smaller percentages of upper transformation products.

United States Steel Co.
Pittsburgh 30, Pa., July 16, 1952

TABLE I.- IMPACT PROPERTIES

						- 1
	Tempering		嵒	Energy absorption (ft-lb)	ion (ft-lb)	
Microstructure	$\begin{array}{c} \texttt{temperature} \\ (^{O}F) \end{array}$	Rockwell C hardness	Room temperature	Average	-40° F	Average
100 percent martensite	800 1000 1200	43.5 37 27	18-18 38-40 71-75	18 39 73	15-15 33-34 60-65	15 33.5 62.5
l percent bainite	800 1000 1200	43 35.5 26	17-18 43-45 68-70	17.5 44 69	12-13 42-46 65-66	12.5 44 65.5
5 percent bainite	800 1000 1200	25.53	16-18 40-42 69-69	1.7 4.1 69	12-13 40-41 68-69	12.5 40.5 68.5
10 percent bainite	800 1000 1200	41 35 23	15-15 37-38 71-72	15 37.5 71.5	11-12 28-31 65-68	29.5 6.5
20 percent bainite	800 1000 1200	39 34.5 22.0	15-17 38-38 74-75	16 38 74.5	11-11 20-25 69-69	11 22.5 69
50 percent bainite	800 1000 1200	35 30.5 20.5	15-15 40-40 68-72	15 40 70	10-11 15-16 33-35	10.5 15.5 34
5 percent ferrite	800 1000 1200	43 37 24	18-20 38-39 73-76	19 38.5 74.5	12-14 33-36 66-68	13 34.5 67
5 percent pearlite (plus 8 percent ferrite)	800 1000 1200	37 34.5 21.5	14-14 31-32 67-68	14 31.5 67.5	10-11 23-23 51-53	10.5 23 52

TABLE II .- TENSILE PROPERTIES

Microstructure	Tempering temperature	Rockwell C		ength osi)	tensile	Elongation (percent	of area
·	(° _F)	nardness	Yield	Tensile	ratio	in l in.)	(percent)
100 percent martensite	800 900 1000 1100 1200	42.5 39 35.5 32 28	193,000 174,000 156,500 134,000 120,000	168,000 149,000	0.95 .93 .93 .90	13.0 14.5 16.5 18.0 21.5	48.0 47.0 53.0 58.0 61.5
100 percent martensite	800 900 1000 1100 1200	45 42 39 34 31		190,500	.94 •95 •96 •93 •92	15.0 16.0 16.0 17.0 20.0	50.5 52.5 51.5 59.0 64.0
1 percent bainite	800 900 1000 1100 1200	44 40.5 37 33 29.5	199,500 183,500 167,000 146,500 131,000	191,500 174,000 155,000	.95 .96 .96 .95 .96	12.5 15.5 17.0 20.0 23.0	51.5 49.5 55.0 59.0 61.5
5 percent bainite	800 900 1000 1100 1200	· 43 39 36 33 29	186,000 169,000 154,500 136,000 118,500	185,000 170,500 154,000	.92 .92 .91 .89	12.5 13.0 14.0 19.0 22.0	51.0 52.0 55.5 58.0 62.0
10 percent bainite	800 900 1000 1100 1200	42 39 36 32 28	177,000 163,500 150,000 130,500 114,500	180,000 166,000 146,500	.90 .91 .90 .89	11.5 15.0 16.5 20.0 23.5	43.0 48.5 52.5 57.0 59.5
20 percent bainite	800 900 1000 1100 1200	41.5 37 35.5 32 27	181,000 168,500 151,000 131,000 113,000	179,500 166,000 148,500	.93 .94 .91 .88 .85	14.5 16.5 17.5 20.0 23.0	52.0 54.5 54.5 58.0 60.5
50 percent bainite	800 900 1000 1100 1200	37 34.5 32 28 26	134,500 126,500 117,500 107,500 97,500	156,500 144,500	.81 .81 .83 ,78	17.0 18.5 19.5 23.0 23.0	47.0 50.0 50.5 56.0 57.0
5 percent ferrite	800 900 1000 1100 1200	40.5 37 34 29.5 24	205,000 186,500 168,000 148,500 122,000	178,000 160,000	.94 .96 .94 .93 .89	13.0 14.5 18.0 19.5 23.5	48.0 50.0 52.5 57.0 62.0
5 percent pearlite (plus 8 percent ferrite)	800 900 1000 1100	41 38.5 36 31.5 29	169,500 159,500 148,000 131,500 119,500	176,500 165,500 149,000	.87 .90 .89 .88 .89	10.5 13.5 16.5 18.5 22.0	40.0 45.5 48.5 54.5 58.0

 $l_{
m These}$ tests were carried out by the Bethlehem Steel Corp.; all other tensile tests were carried out by the laboratories specified in the text.

TABLE III.- FATIGUE PROPERTIES

Microstructure	Tempering temperature (°F)	Rockwell C hardness	Endurance limit (psi)	Endurance-limit and tensile- strength ratio	Endurance- limit and yield-point ratio
100 percent martensite	800 900 1000 1100 1200	42.5 39 35.5 32 28	79,000 77,500 76,000 70,500 65,000	0.39 .42 .45 .47 .49	0.41 .45 .49 .53
l percent bainite	800 900 1000 1100 1200	43 40 37•5 34 28•5	79,000 73,000 72,000 67,000 65,000	.37 .39 .41 .43 .48	.40 .40 .43 .46 .50
5 percent bainite	800 900 1000 1100 1200	43.5 39 37 32.5 28	82,000 77,000 75,000 72,000 64,000	.40 .42 .44 .47 .47	.44 .46 .49 .53 .54
10 percent bainite	800 900 1000 1100 1200	42 39 36 31 27	81,000 79,000 74,000 69,000 68,000	.41 .44 .45 .47 .52	.46 .48 .49 .53 .59
20 percent bainite	800 900 1000 1100	41 36.5 34.5 31.0 26.0	83,000 75,000 68,000 66,000 62,000	.43 .42 .41 .44	.46 .44 .45 .50 .55
50 percent bainite	⁸ 800 800 ⁸ 800 900 1000 1100	39.5 33.5 34 32.5 29.5 25.5 24	72,000 56,000 47,000 59,000 57,000 54,000 49,000	 •33 •28 •38 •39 •42 •39	.44 .35 .47 .49 .50
5 percent ferrite	800 900 1000 1100 1200	40.5 37.0 34.0 29.5 24.0	87,000 85,000 84,000 70,000 61,000	.40 .44 .47 .44 .46	.42 .46 .50 .47 .50
5 percent pearlite (plus 8 per- cent ferrite)	800 900 1000 1100 1200	43 39.5 37.5 32.5 29.5	67,000 66,000 68,000 62,000 57,000	.34 .37 .41 .42 .42	.40 .41 .46 .47 .48

^aDouble tempered. ^bRecheck.

TABLE IV. - RATINGS1 OF MECHANICAL PROPERTIES

Microstructure	Impact strength at room temperature	Impact strength at -40° F	Yield strength	Elongation	Reduction of area	Endurance limit
T. M. + 1 percent bainite	100	700	100	100	100	100
T. M. + 5 percent bainite	92	66	16	100	100	100
T. M. + 10 percent bainite	82	87	16	100	100	100
T. M. + 20 percent bainite	80	29	97	100	100	100
T. M. + 50 percent bainite	30	50	4.8	100	95	87
T. M. + 5 percent ferrite	100	100	100	100	100	100
T. M. + 5 percent pearlite	19	57	. 93	100	95	06

Expressed as percentage of tempered-martensite (T. M.) value at constant hardness of 3^4 Rockwell C.

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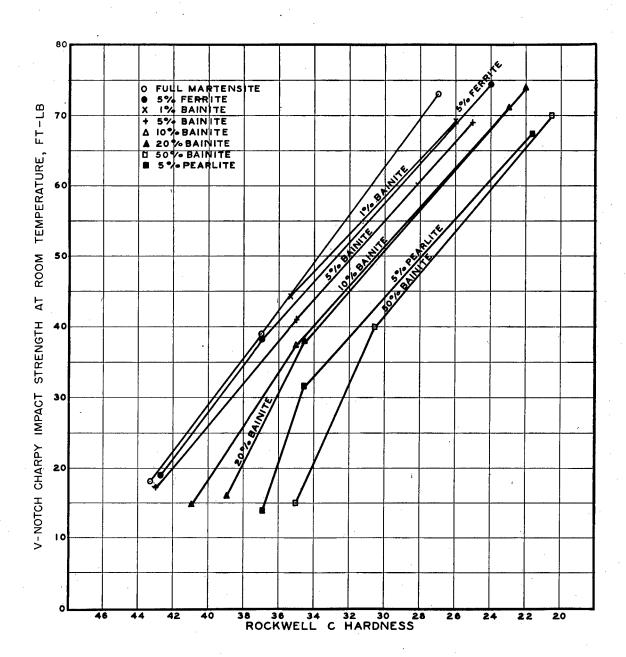


Figure 1.- Effect of nonmartensitic products on impact strength at room temperature.

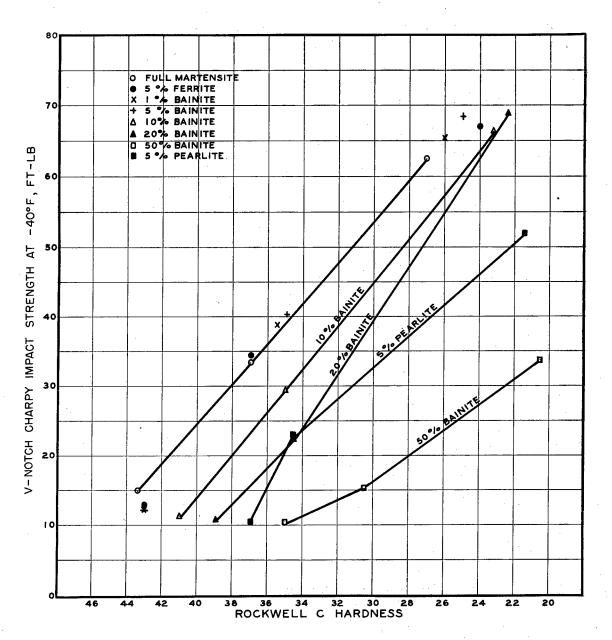


Figure 2.- Effect of nonmartensitic products on impact strength at -40° F.

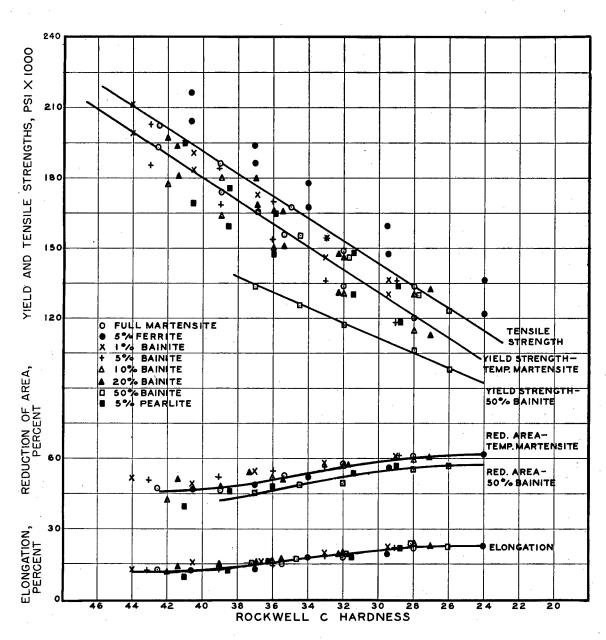


Figure 3.- Effect of nonmartensitic products on tensile properties.

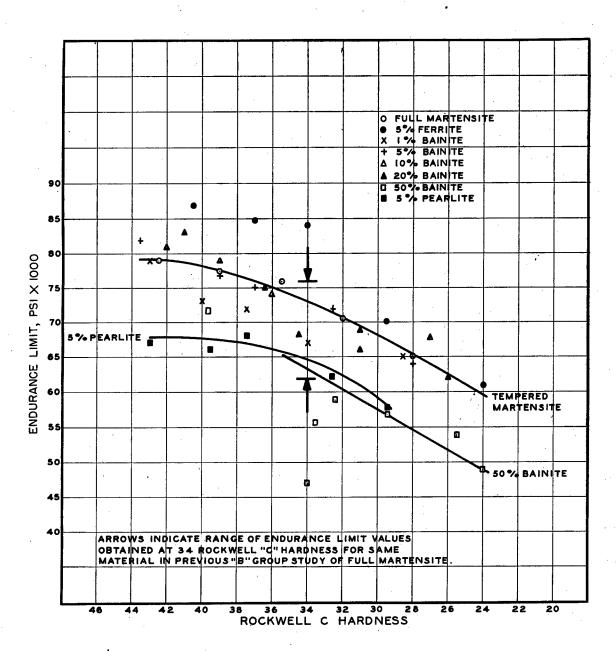


Figure 4.- Effect of nonmartensitic products on endurance limit.

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Figure 5.- Effect of upper bainite on mechanical properties at 34 Rockwell C.

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